

Gamma Irradiation Effects on Physical, Optical, Structural and Radiation Shielding Properties of Tellurite based Glasses

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Abstract

Tellurite glasses have been researched for their radiation shielding properties as a potential alternative to lead and lead silica glass, which poses toxicity concerns. The effects of radiation on tellurite glasses are assessed using both physical irradiation and simulation with the Phy-X/PSD software. Glasses with the composition $(70-x-y) \text{TeO}_2-20\text{ZnO}-9\text{Na}_2\text{O}-1\text{Er}_2\text{O}_3-(x)\text{TiO}_2-(y)\text{Al}_2\text{O}_3$, were fabricated using the melt-quench method. These glasses were then irradiated with gamma radiation at different doses. Characterization techniques, including X-ray diffraction (XRD) and UV-VIS spectroscopy, along with density measurements, were applied to the glasses both before and after irradiation. The XRD results confirmed that the glass samples were amorphous. UV-VIS spectroscopy showed that transmittance decreased as the radiation dose increased. The Phy-X/PSD simulation program was used to model the radiation properties of the glasses based on their dosage and composition. The simulation results indicated that the half-value layer (HVL) and mean free path (MFP) increased post-irradiation and then remained constant. These findings suggest that tellurite glasses, with their enhanced radiation shielding properties, could be a viable, safer alternative to lead-based glasses for various applications.

Keywords: Tellurite Glass, Titanium NPs, Aluminium oxide NPs, gamma, Phy-X/PSD Simulation

1. Introduction

Ever since the advent of nuclear technology, the need for radiation-resistant glasses has grown steadily, driven by the expanding development and presence of radiation sources. The increase in space exploration has further necessitated the production of these specialized glasses [1]. In space, the absence of an atmospheric layer to protect astronauts and their equipment from radiation makes the problem even more pronounced, as all forms of radiation can easily penetrate materials without such protection.

Additionally, the field of medical technology has seen significant growth in the use of radiation for both treatment and diagnostic purposes, underscoring the need for advanced radiation-resistant materials. These applications demand materials that can effectively shield against harmful radiation while maintaining their structural integrity and optical properties. As such, the development of radiation-resistant glasses is crucial for ensuring safety and performance in a wide range of environments, from space missions to medical facilities [2].

As a result of the wide use of radiation technology, numerous methods of resisting radiation for a human user in these circumstances have been developed. In the medical field, lead aprons are often used to shield the patient and medical personnel alike when conducting radiation-based scanning of the body, such as x-ray and MRI; as well as when receiving radioactive medicine, such as radiotherapy. In nuclear reactors, thick walls of concrete are constructed around the reactor, along with the administration and control buildings having extra thick walls. Spacecraft also use metal casings to shield the components and the astronauts from unfiltered radiation from space. All these methods leave little to no room for visibility through the shielding [3]. Thus, the material's radiation sensitivity must be thoroughly researched before it is subjected to a radiation environment [4].

Hence, a glass with radiation-resistant properties and being transparent enough to look through in addition to ease of manufacturing has been highly requested [5]. To date, silicate-based glasses lined with lead oxides have been used for radiation protection. Lead glasses are common in usage; however, the lead content has led to a reduction in the melting point and hardness of the glass. In addition to the environmental issues that result from the use of lead, there is a need for alternatives to be developed [6]. On that account, a prominent element for this research has been Tellurite. Tellurite exhibits desirable chemical and physical properties such as low melting point, low phonon energy, low crystallization ability, high dielectric constant, and high thermal stability [7]. There have been several investigations into tellurite glass compositions, such as erbium-doped tellurite glass [8], [9], tellurite glass with zinc fluoride nanoparticles [10], and titania-bismuth-borotellurite glass [11]. In particular, there is interest in the development of compositions containing titania and aluminium oxide

nanoparticles [12]–[15]. Furthermore, radiation damage to the tellurite glass system has received significant attention in recent studies for radiation shielding applications [16].

In the study, glass samples were discovered to be transparent based on optical properties [17]. It is light yellow with a concentration of 10 mol% TeO₂. It began to obtain a much darker colour with rising TeO₂ concentration before it became dark red [17]. In another study the tellurite glass system at ambient temperature, a ⁶⁰Co gamma cell (2000 Ci) was employed as a gamma-ray source with a dosage rate of 1.5 Gy s⁻¹ (150 rad s⁻¹). After irradiation, the colour of a glass system changed from transparent yellow to brown [18]. The acquired spectra reveal that all edges display a redshift as the radiation dose increases. The level of resistance to gamma-irradiation depends on the redox potential of the individual ions [18].

Furthermore, alternative methods to test the effects of radiation are being researched, which have the potential to replace direct testing using actual radiation. These methods include simulation by computer programs such as Phy-X/PSD, among others. For the purposes of this research, the simulation program Phy-X/PSD is used. Phy-X/PSD is a web-based simulation program easily accessible at the address of <https://phy-x.net/PSD>, developed by Sakar et al. in 2020. The program is capable of simulating a range of radiation energies, as well as some well-known radioisotopes' energies. The number of types of material being simulated is also virtually infinite as the input can be any combination of the periodic table and its corresponding density.

Currently, Phy-X/PSD is rather new to the scene of radiation and particle simulation programs. Thus, the study on its suitability and its usage in research is currently ongoing. It has been determined that Phy-X/PSD can be used as one of the theoretical approaches to calculate the photon attenuation factors [3]. Other than that, it is noted that the program is fast and accurate in their calculations. The computation of the parameters is performed by the database through the chemical formula and the density of the glass specimens [5]. The output of these calculations can then be saved as an Excel file by the user.

In this work, the tellurite glass with a composition of $(70-x-y) \text{TeO}_2-20\text{ZnO}-9\text{Na}_2\text{O}-1\text{Er}_2\text{O}_3-(x)\text{TiO}_2-(y)\text{Al}_2\text{O}_3$, with a total mass of 20 g per glass sample was fabricated. Analysis was then performed on the physical, structural, and optical properties pre- and post-irradiation, along with simulating the radiation analysis using Phy-X/PSD. The irradiation is with total ionizing doses (TID) of 5, 10, 50, 75, and 100 kGy separately to each sample.

2. Experimental Procedures

2.1. Fabrication of Glass

In this project, two series of samples were prepared by conventional melt quenching process. The raw materials obtained from Sigma Aldrich, included tellurium dioxide (TeO_2 , purity 99%), zinc oxide (ZnO , purity $\geq 99\%$), sodium oxide (Na_2O , purity 80%), erbium III oxides (Er_2O_3 , purity $>99.99\%$), titanium IV oxide (TiO_2 , with a primary particle size of 21nm, purity $\geq 99.5\%$), and aluminium oxide nanopowder (Al_2O_3 , with a primary particle size of 13 nm, purity 99.8%) with a total mass of 20 g per glass sample. The samples were prepared with the compositions of $(70-x-y) \text{TeO}_2.20\text{ZnO}.9\text{Na}_2\text{O}.1\text{Er}_2\text{O}_3.(x)\text{TiO}_2.(y)\text{Al}_2\text{O}_3$, where the sample with code TZNETA3 having $x = 0.3$ mol% and $y = 0.6$ mol% and sample with code TZNETA6 having $x = 0.6$ mol% and $y = 0.3$ mol%. The composition masses are listed in Table 1. A platinum crucible containing the materials were prepared and melted in an electrical furnace at 950°C for 30 min. The mixture was then poured into a brass mould and held at 250°C for annealing for 3 h.

The samples were then characterized before and after irradiation. The samples were sent to Malaysia Nuclear Agency, with the specific facility of MINTec-SINAGAMA Irradiation Plant. This facility uses ionizing energy in the form of gamma radiation with Cobalt-60 as its source. The samples were brought to the plant for irradiation, specifically utilizing the JS10000 IR 219 tote irradiator system with an irradiation rate of 2 kGy/h. Then, the samples were irradiated with TID of 5, 10, 50, 75, and 100 kGy, each dose applied separately to individual samples, using an average gamma energy of 1.25 MeV. The ceric-cerous dosimeter was used to analyze the radiation dose absorbed by all samples, and the Total Ionizing Dose (TID) in these studies was verified based on the evaluation of the ceric-cerous system for gamma radiation exposure. This dosimetry system operates based on the oxidation-reduction

reaction between ceric (Ce^{4+}) and cerous (Ce^{3+}) ions in an acidic solution when exposed to ionizing radiation. The radiation-induced reduction of Ce^{4+} to Ce^{3+} results in a measurable change in optical absorbance, typically monitored using spectrophotometry. The glass samples were exposed to Co-60 gamma radiation (1.25 MeV) at various TID, and structural and optical changes were analyzed before and after irradiation.

Table 1 Chemical composition of glass samples.

Glass code	TZNETA3		TZNETA6	
	(mol%)	Mass (g)	(mol%)	Mass (g)
TeO ₂	69.1	16.122	69.1	16.129
ZnO	20	2.379	20	2.380
Na ₂ O	9	0.815	9	0.816
Er ₂ O ₃	1	0.559	1	0.559
TiO ₂	0.3	0.035	0.6	0.070
Al ₂ O ₃	0.6	0.089	0.3	0.045
Total	100	20	100	20

The density of the glass samples with 2 sample each for pre- and post-irradiation was measured. The density of glass was determined using an analytical balance of specific density-PrecisaXT220A. The density of glasses was measured using the suspended weight method based on Archimedes' principle at room temperature and computed using equation (1). The calculation for the density of the samples is then expressed as:

$$\rho = \frac{w}{|w_d|}(\rho_0 - d) + d \quad (1)$$

Where, ρ is the density of the sample, W is the weight of sample in air, W_d is then difference between weight of sample in air and in water, ρ_0 is the density of water at 29 °C (0.99651 g/cm³), and d is a constant.

Model Lambda UV-VIS spectroscopy EZ210 was used to achieve UV-VIS spectroscopy of the proposed glasses. Optical transmittance of the glass samples was measured both before and after irradiation. Using these transmittance measurements in the UV and visible spectral ranges, the optical energy gap and radiation-induced absorption can be calculated [19].

Bruker D2 Phaser was used to achieve XRD analysis of the proposed glasses. After which, the data is extracted to acquire the intensity peak graph. Corresponding peaks in the graph were then identified for their respective structures [20].

2.2. Radiation Shielding Properties

PSD, standing for Photon Shielding and Dosimetry is a simulation software made readily available online at the website of <https://phy-x.net/PSD>. This software is developed for the calculations of parameters related to shielding and dosimetry. Some of these parameters include mass attenuation coefficient (MAC), half value layer (HVL), effective atomic number (Z_{eff}), and energy absorption buildup factor (EABF). The software is capable of generating data on the shielding parameters in the continuous energy region of 1 keV to 100 GeV, as well as some well-known radioactive sources such as Cobalt-60, Caesium-137, Sodium-22, and Iodine-131. The energy ranges to be tested can be selected by the user at will depending on their research requirements. The material being tested can also be inserted by the user by inputting the composition and the density of the material.

Phy-X/PSD runs on a remote server that has Intel® Core™ i7-2600 CPU @ 3.40 GHz CPU with 1 GB installed memory and its operating system being Ubuntu 14.03.3 LTS. The application language is NodeJS v8.4.0 serving with Nginx 1.15.8. Security between client's browser and the server is being established with 256 Bit Positive SSL [21].

Direct measurement of gamma-ray absorption in glass requires specialized instruments such as gamma spectroscopy, which may not always be available. Instead, this study used the Phy-X/PSD software, a well-validated technique for computing shielding parameters based on the glass's elemental composition and density. The software calculates important shielding parameters such as Mass Attenuation Coefficient (MAC), which measures the likelihood of gamma photon interactions with the material. Half-Value Layer (HVL) is the thickness necessary to reduce gamma intensity in half, while Mean Free Path (MFP) is the average distance gamma rays travel before interacting with the material. These measurements provide an indirect but reliable assessment of the glass's efficiency as a gamma shield.

2.2.1. Mass attenuation coefficient (MAC)

The mass attenuation coefficient is defined as the probability per unit mass that a photon will interact with the material through absorption or scattering. Thus, this number is often presented as a fraction. The MAC can be derived from the Beer-Lambert law, which is the relationship between the attenuation of light through a substance and the properties of that substance [22]. The relationship can be expressed in the form as Equation (2) [23]:

$$I = I_0 e^{-\mu x} \quad (2)$$

Where, I is the incident intensity, I_0 is the transmitted intensity, μ is the linear mass coefficient, x is the thickness.

Which we can then use to calculate the mass attenuation coefficient by using the relation of Equation (3) [23]:

$$\mu_m = \frac{\mu}{\rho} \quad (3)$$

Where, μ_m is the mass attenuation coefficient and ρ is the density

When used with MFP and HVL, the relations then become Equations (4) and (5) [23]:

$$MFP = \frac{1}{\mu} \quad (4)$$

$$HVL = \frac{\ln 2}{\mu} \quad (5)$$

2.2.2. Mean free path (MFP)

The mean free path is defined as the average distance traveled by a photon before an interaction takes place, such as changing direction, and change in velocity. From this relation, we can say that the lower the MFP value, the better the resistance against radiation. The photon loses energy as it interacts with the medium, therefore reducing its penetrating power as well.

2.2.3. Half-value layer (HVL)

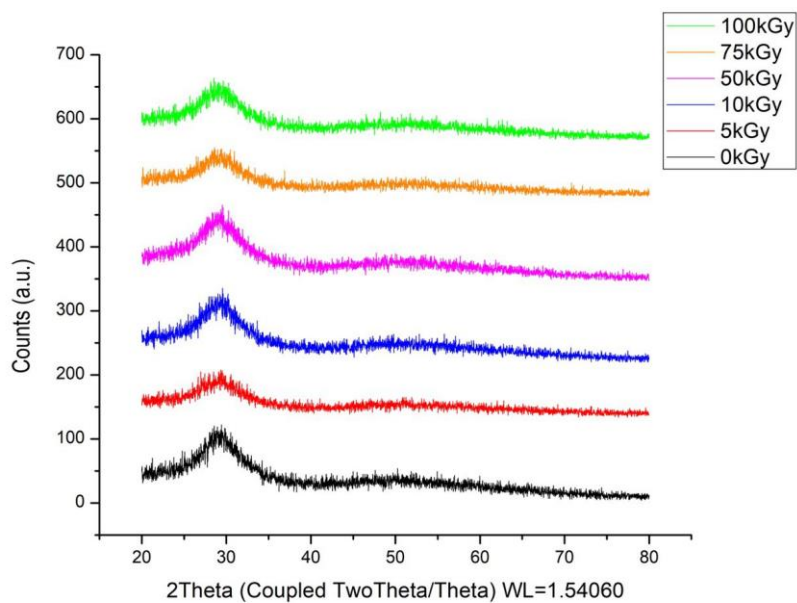
The half value layer is defined as the thickness of the material where half the intensity of the incident photon is maintained [23]. In simpler terms, HVL is the length of the material at which the penetrating photon reduces its intensity by half [23]. This indicates that the lower the HVL value, the better the resistance against radiation. As the energy of the photon is halved by an increasingly thinner material, the penetrating power of the photon when it exits the material reduces greatly as well. This means that we can fabricate a thinner glass, thus saving on material and cost, while providing sufficient radiation shielding.

4. Results and discussion

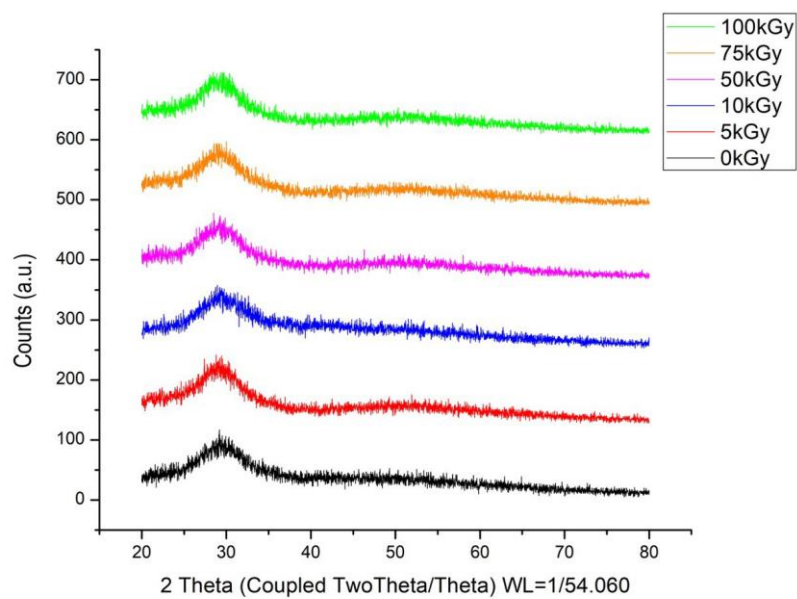
4.1. Structural properties

4.1.1. XRD

The samples were crushed finely using an agate mortar and pestle and packed in a conical tube. The amorphousness of the glass is analysed using XRD. Fig 1 shows the XRD spectra of glass with varying dosages of gamma radiation between samples TZNETA3 and TZNETA6 respectively. The appearance of broad diffuse scattering located at the 25–35 degrees with the absence of sharp crystallization peaks signifies the amorphous nature and indicates the short-range order of atoms in the glass [12]. The XRD patterns recorded are identical based on Fig 1 before and after gamma ray radiation. As a result, the XRD patterns confirm the existence of an amorphous glass system [24].



(a)



(b)

Fig. 1. XRD spectra of (a) TZNETA3 and (b) TZNETA6

4.1.2. Density

The density for TZNETA3 decreases immediately once irradiated. The data has slight changes between 5.13396 gcm^{-3} to 5.14515 gcm^{-3} post-irradiation as shown in Fig 2 and detailed in Table 2 in

Appendix A. This indicates that the damage done to the glass is immediate and constant regardless of the amount of dosage applied. This is due to the energy being constant through all dosages, which is at 1.25 MeV. When bonds between oxygen and network-forming atoms break, the oxygen atoms may not make new links immediately. These unbound oxygen atoms, known as non-bridging oxygens (NBOs), remain within the glass structure. NBOs can affect the physical qualities of glass, such as density. Density changes occur as atoms rearrange and voids emerge, altering the material's compactness.

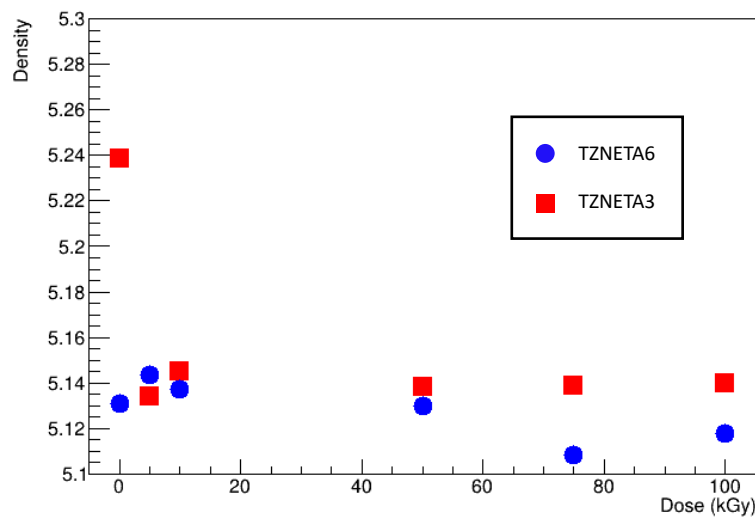


Fig. 2. Density of TZNETA3 and TZNETA6 pre- and post-irradiation

Table 2 Density of samples.

Dosage (kGy)	TZNETA3	TZNETA6
Not Irradiated (0)	5.21758	5.13083
5	5.13396	5.14365
10	5.14515	5.13737
50	5.13814	5.12997
75	5.13875	5.10825
100	5.14008	5.11774

The dosage does not affect the penetrative power of the radiation, which is only affected by the irradiation energy. A study found that the penetrating power is correlated with the irradiation energy, with a weak energy stopping within the sample and a strong energy penetrating the sample entirely [25]. Since the energy is constant throughout the irradiation process at 1.25 MeV, the penetration of the radiation on the glass samples is constant and the damage done is on the same layers on all samples. Thus, the density remains constant post-irradiation.

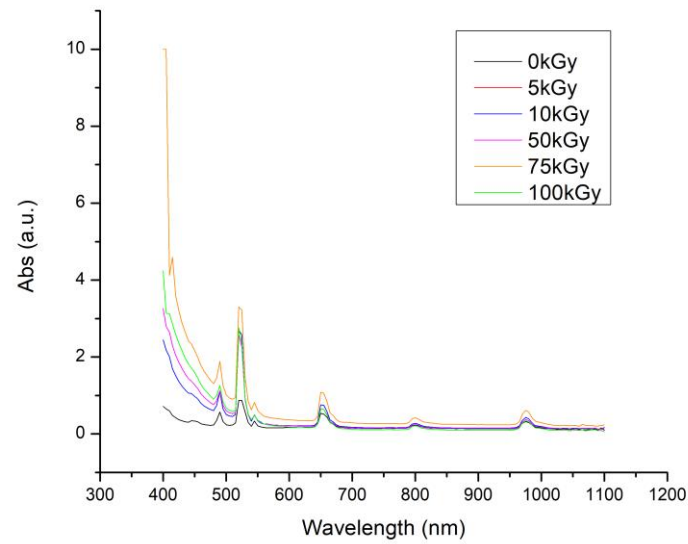
The initial trend for the TZNETA6 sample increases instead of decreasing, while the post-irradiation readings vary slightly between 5.10825 gcm^{-3} and 5.14365 gcm^{-3} . This is due to the outliers in the density of the glass pre-irradiation, where the average is taken for comparison pre- and post-irradiation analysis. These minor differences are important indicators that the structural damage is mostly caused by void creation and atomic reconfigurations.

As the dose continues to increase, there is a very slight decrease in the density of the glass. This is attributed to the expansion that occurs in the glass, which comes from the formation of voids [26]. According to Ruller and Friebele in 1991 [27], the voids are produced when there is trapping of electrons or holes formed by ionization in the network. As a result, it will cause the bond to rupture or increase in length. The formation of voids is a complex process that must take into account many factors, where in this case, the chemical composition and the microstructure of the material. The samples investigated here created voids by the ionization of the atoms, which breaks the bonds between the atoms and causes it to be pushed around into the interstitial space. When enough atoms experience this effect, the space left behind creates a void.

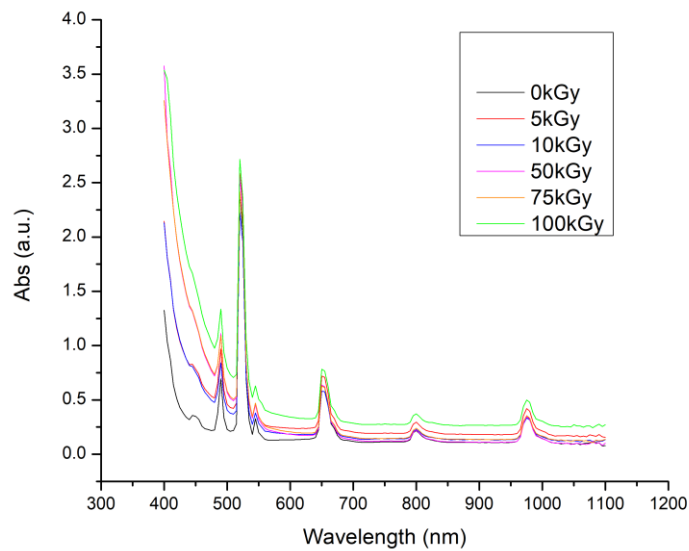
4.2. Optical properties

The optical properties in the spectra range of 400-1000 nm were analysed by the ultraviolet-visible (UV-VIS) spectroscopy using Model Lambda EZ210. Fig 3 and 4 show the UV-VIS absorption and transmission spectra of the glass samples of TZNETA6 and TZNETA3. Generally, the spectral features of both sample series were found to be qualitatively similar. The absorption peaks were

attributed to structural rearrangements of the glass due to the 4f-4f transition in the erbium ions in the glass matrix [28].

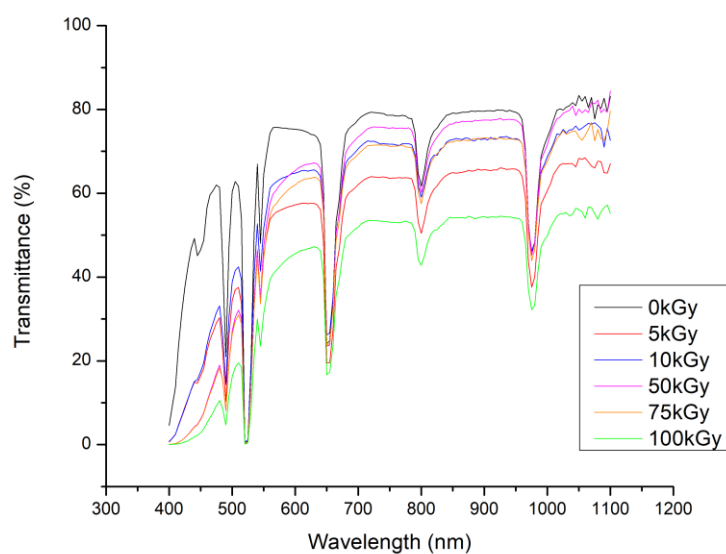


(a)

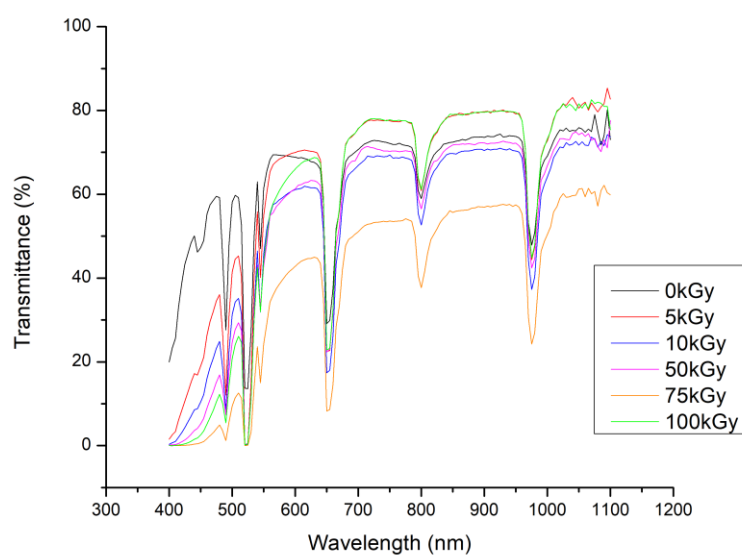


(b)

Fig. 3. UV-VIS absorption spectra for (a) TZNETA3 and (b) TZNETA6



(a)



(b)

Fig. 4. UV-VIS transmittance spectra of (a) TZNETA3 and (b) TZNETA6

Seven absorption peaks were observed centered at different wavelengths and their corresponding erbium transitions are summarized in Table 3. The transition of the erbium ions was tabulated based on energy value referring to energy levels of trivalent lanthanide aquo ions [29].

Table 3 Absorption peaks centre and corresponding Erbium transitions

Absorption band centre (nm)	Erbium transition energy level
445	$^4F_{3/2}$
490	$^4F_{7/2}$
519	$^2H_{11/2}$
545	$^4S_{3/2}$
650	$^4F_{9/2}$
800	$^4I_{9/2}$
975	$^4I_{11/2}$

The absorption spectra consist of seven absorption bands centred around the wavelengths of 445, 490, 519, 545, 650, 800, and 975 nm. These peaks represent the transition of erbium ions from ground state $^4I_{15/2}$ to excited state energy levels of $^4F_{3/2}$, $^4F_{7/2}$, $^2H_{11/2}$, $^4S_{3/2}$, $^4F_{9/2}$, $^4I_{9/2}$, and $^4I_{11/2}$ respectively [8].

A study found additional absorption bands at 554 nm and 827 nm corresponding to the Ti and Al atoms, respectively. However, these weak bands reflect the inconsistent symmetry in the distribution of the nanoparticles (NPs). The absence of these bands in the present data is also due to the small resolution of the acquired data, that the bands are insignificant when compared to the intensity of the peaks made by the erbium ions [12]. The bands that are acquired in this study are done by fabricating a separate series of samples that excludes Er_2O_3 from the composition, thus the bands are clearer than that with erbium ions.

In the transmittance spectra for TZNETA6 sample, there is a decrease in the transmittance as the dosage increases, with the exception of 75 kGy being the lowest. Meanwhile, the transmission spectra for TZNETA3 sample is also decreasing except for the 5 kGy sample. As the energy increases, the transmittance of the glass decreases due to the damage done. According to the research by [30] the effects of gamma radiation effects on optical properties on glass with a composition of $CeO-Na_2O-$

SrO–B₂O₃ is investigated. The data acquired here corresponds to the findings of [30] in that the transmission of the glass samples were decreased after exposure to gamma irradiation in all wavelengths.

The damage done to the glass is caused by the ionization of the atoms in the glass, which comes from 3 types of interactions of gamma rays with the atoms: the photoelectric effect, the Compton effect, and pair production. These contribute to the calculation of the total absorption coefficient [31]. Additionally, the ionization of the atoms also causes the formation of non-bridging oxygens (NBO). This process causes the chemical bonds to break, releasing oxygen atoms and ions, which then enters the interstitial spaces in the glass network, becoming interstitial atoms. The atoms then become bonded to the nearest atoms, which is the tellurite network, causing the TeO₃ units to change into TeO₄ trigonal bipyramids, which contribute to the alteration of the other properties of the glass [32].

The alterations to the glass structure and network causes the optical properties to change as well. The formation of NBO's increases the interaction of the light rays entering the glass sample. This then causes the absorption spectra to increase as the higher the dosage, the more the NBO's are formed due to the increasing damage. The increase in absorption can also be due to the generation of additional erbium ions from irradiation. When a glass containing a transition or rare earth ions is irradiated, the ions act as a trap for released electrons and holes during the irradiation process [33]. Thus, in general, we can say that the higher the energy, the higher the interaction with atoms, and the higher the absorption.

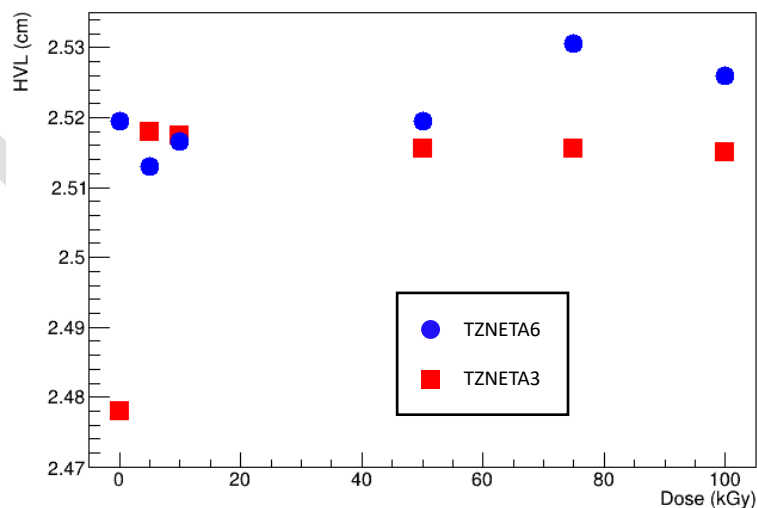
Other defects are also found as an effect from the irradiation. The generation of O- centres after gamma irradiation causes to many additional absorption effects. These effects come in the form of additional absorption spectra and peaks at certain wavelengths [33]. It is then concluded that the overlap of multiple damages and defects contributes to the increase of amplitude and bandwidth in absorption, thereby decreasing the transmittance of the glass sample.

The damage from the irradiation is also found to be permanent and irreversible. This is due to the non-conducting nature of the material, where mobility of the defects is very small or negligible.

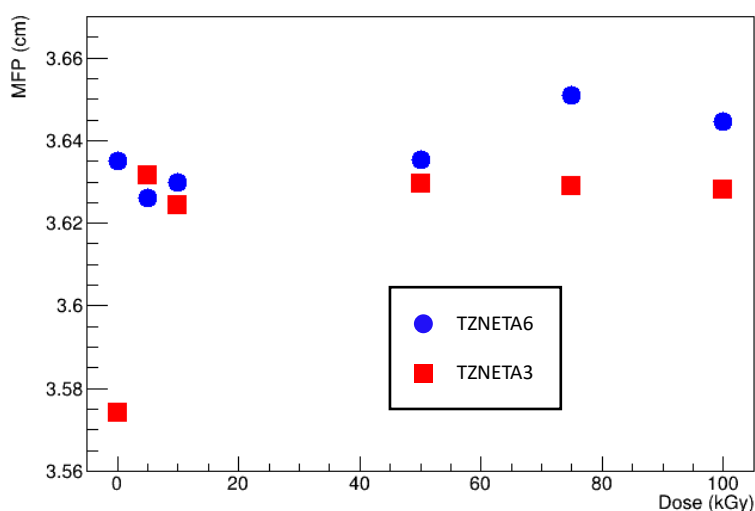
4.3. Radiation properties

The trend for HVL and MFP of the samples pre- and post-irradiation can be seen in Fig 5. In general, the values for HVL and MFP increase as the sample is irradiated and then is constant through all the dosages. This is due to the decrease in density as irradiation increases. It is noted that the HVL and MFP are inversely proportional to the density of the sample. In particular, the sample TZNETA3 increases its HVL value from 2.4775 to 2.5175 cm at 5 kGy dose irradiation, while the sample TZNETA6 initially decreases its HVL value from 2.5195 to 2.5130 cm at 5 kGy dose irradiation but then increases again to its highest point of 2.5305 cm at 75 kGy dose irradiation. The MFP shows a similar trend in that the value for TZNETA3 sample increases from 3.5740 to 3.6315 cm at 5 kGy dose irradiation, while TZNETA6 sample initially decreases from 3.6350 to 3.6260 cm at 5 kGy dose irradiation but then increases again to its highest point of 3.6510 cm at 75 kGy dose irradiation.

The values also remain relatively constant for all readings post-irradiation, with the range between values being 0.0025 cm and 0.0175 cm in HVL, along with 0.0070 cm and 0.0250 cm in MFP for TZNETA3 and TZNETA6 respectively. This trend is not affected by the dosage but instead is due to the constant penetrative power as energy was constant during irradiation.



(a)



(b)

Fig. 5. Graph of (a) HVL and (b) MFP TZNETA3 and TZNETA6 pre- and post-irradiation

The MAC of all samples are observed to be the same. The MAC is determined by the theoretical composition of the sample only. Thus, any changes induced from the irradiation are irrelevant in this case. Table 4 lists the mass attenuation coefficients (MAC), half value layer (HVL), and mean free path (MFP) of the glass samples against the increasing radiation dosage by result of the Phy-X/PSD simulation.

Table 4 MAC, HVL, and MFP readings from Phy-X/PSD of TZNETA3 and TZNETA6 pre- and post-irradiation.

Dosage (kGy)	MAC		HVL (cm)		MFP (cm)	
	TZNETA3	TZNETA6	TZNETA3	TZNETA6	TZNETA3	TZNETA6
Not Irradiated (0)	0.054	0.054	2.4775	2.5195	3.5740	3.6350
5	0.054	0.054	2.5175	2.5130	3.6315	3.6260

10	0.054	0.054	2.5175	2.5165	3.6245	3.6300
50	0.054	0.054	2.5155	2.5195	3.6295	3.6355
75	0.054	0.054	2.5155	2.5305	3.6290	3.6510
100	0.054	0.054	2.5150	2.5260	3.6280	3.6445

In comparison, the density of the glasses influences properties such as HVL, TVL, and MFP, whereas the MAC value has direct connections with experimental results across a range of TID. While direct gamma absorption measurements are ideal, a material's shielding performance is mostly determined by its composition and density, which were thoroughly tested in this study. The agreement between theoretical (Phy-X/PSD) and experimental (irradiation-induced alterations) results confirms the glass's radiation shielding capability. The investigated glasses have a higher MAC than tellurite glasses containing Sm_2O_3 and Nd_2O_3 [34] and chalcogenide glasses [35]. The photon shielding characteristics of the glasses improve with increased NP content. This study demonstrates that the examined glasses can be used as radiation shielding materials even under high TID gamma irradiation.

5.0 Conclusion

This study examined the physical, optical analysis, and radiation characteristics of glass through both physical experimentation and software simulation, exploring the correlation between these methods. Observations showed that the physical properties of the glass remained relatively stable across all samples immediately post-irradiation. However, post-irradiation optical analysis revealed a decreasing trend with increasing radiation dosage, attributed to cumulative radiation damage. In contrast, the radiation analysis indicated that the trend increased immediately post-irradiation and then stabilized, highlighting a discrepancy between the optical and radiation responses due to the differing nature of damage accumulation and energy variation. The glass's shielding efficiency is effectively evaluated using validated computational methods (Phy-X/PSD) and experimental irradiation studies. The results show that the glass composition and density provide adequate gamma-ray attenuation. The findings suggest that continued investigation of tellurite glass containing NPs as a viable alternative for radiation

shielding materials. Additionally, the absence of toxic materials in tellurite glass enhances its appeal. Utilizing simulation programs for material testing against radiation proves advantageous, as it minimizes radiation exposure and allows comprehensive monitoring of variables and results. Nevertheless, further testing is essential to validate the accuracy of the simulation program's results.

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